



## Influences of working pressure on properties for TiO<sub>2</sub> films deposited by DC pulse magnetron sputtering

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### Abstract

TiO<sub>2</sub> films were deposited at room temperature by DC pulse magnetron sputtering system. The crystalline structures, morphological features and photocatalytic activity of TiO<sub>2</sub> films were systematically investigated by X-ray diffraction (XRD), atomic force microscopy (AFM) and ultraviolet spectrophotometer, respectively. The results indicated that working pressure was the key deposition parameter influencing the TiO<sub>2</sub> film phase composition at room temperature, which directly affected its photocatalytic activity. With increasing working pressure, the target self-bias decreases monotonously. Therefore, low temperature TiO<sub>2</sub> phase (anatase) could be deposited with high working pressure. The anatase TiO<sub>2</sub> films deposited with 1.4 Pa working pressure displayed the highest photocatalytic activity by the decomposition of Methyl Orange solution, which the degradation rate reached the maximum (35%) after irradiation by ultraviolet light for 1 h.

**Key words:** TiO<sub>2</sub> film; anatase; UV induced photocatalysis; DC pulse magnetron sputtering

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### Introduction

Over the last few years, a great attention has been focused on the titania (TiO<sub>2</sub>) films due to its excellent chemical stability, high refractive index, nontoxicity, and good mechanical properties. Besides, the TiO<sub>2</sub> films can exhibit excellent photocatalytic and superhydrophilic properties after ultraviolet (UV) light irradiation (Doong *et al.*, 2007; Zhao *et al.*, 2008; Miyamura *et al.*, 2008; Wang *et al.*, 2008; Turkevych *et al.*, 2008). Therefore, TiO<sub>2</sub> films can be utilized in many applications, such as antifogging, selfcleaning, antibacterial, self-sterilization processes and in the removal of organic pollutants from wastewater (Aubry *et al.*, 2007; Tavares *et al.*, 2007). But there are several drawbacks that limit a wider utilization of the TiO<sub>2</sub> films as photocatalyst applications. The most primary one is the formation of photoactive TiO<sub>2</sub> films at room temperature without post-deposition thermal annealing treatment (Heft *et al.*, 2006).

There are three different crystalline phases of TiO<sub>2</sub> namely anatase, rutile and brookite. Among them, the anatase structure is referred to be the most photoactive structure. However, the deposition of crystalline photoactive TiO<sub>2</sub> films without the substrate heating or a post-deposition thermal annealing has not been fully mastered yet. Among numerous deposition methods, the magnetron sputtering is a promising one for the deposition

of crystalline anatase TiO<sub>2</sub> films with photocatalytic at room temperature (lower than 100°C).

In this article, anatase phase TiO<sub>2</sub> films are deposited with direct current pulse magnetron sputtering system. This method uses high purity Ar (99.999%) as sputtering gas and high purity O<sub>2</sub> (99.999%) as reactive gas. High purity Titanium (Ti, 99.99%) is used as target material. By this method, the anatase TiO<sub>2</sub> films with excellent photocatalytic are deposited at room temperature.

### 1 Experiment

#### 1.1 Sample collection

A Ti (99.99%) target (200 × 100 mm<sup>2</sup>) was used for a reactive sputtering material for depositing transparent TiO<sub>2</sub> films in the Ar + O<sub>2</sub> mixture. The target power was supplied by a DC pulse (Sicha *et al.*, 2008). Power supply unit operating in power mode was at a repetition frequency of 100 kHz and a duty cycle of 0.1 (Pinnacle Plus + 5 kW, Advanced Energy Industries Inc., USA). Films were deposited on unheated glass substrates. The distance between substrate and target was 110 mm. The substrate surface temperature (*T*) was measured by thermocouple and *T* was lower than 100°C in all experiments. In this case, the contamination of TiO<sub>2</sub> films by the Na<sup>+</sup> diffusing from the glass substrates can be neglected. The influence of working gas pressure on the TiO<sub>2</sub> film structure and photocatalytic activity was investigated. The detailed deposition

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conditions are listed in Table 1.

The film phase composition was determined by the X-ray diffraction (XRD) analysis using a D/max2500 diffractometer (Rigaku, Japan) working in Bragg-Brentano geometry with Cu  $K_{\alpha}$  (40 kV, 40 mA) radiation. The morphological features of films were measured by atomic force microscopy (AFM) system (SPM-9500J3, Shimadzu Corporation, Japan). The thickness of films was measured by a stylus profilometer Dektak 6M (Veeco, USA) with the resolution of 0.1 nm. The photocatalytic activity of TiO<sub>2</sub> film was determined from a decomposition of C<sub>14</sub>H<sub>14</sub>N<sub>3</sub>NaO<sub>3</sub>S (Methyl Orange, MO) solution. This solution exhibits a very good stability against UV light irradiation. The TiO<sub>2</sub> films were immersed in the MO solution with initial concentration ( $C_0 = 10$  mg/L) in distilled water (10 mL) and irradiated by the UV light for 1 h. The concentration of solution was determined by measuring the absorbency of solution at wavelength 485 nm using ultraviolet-visible spectrophotometer (U-3310, Hitachi Ltd., Japan).

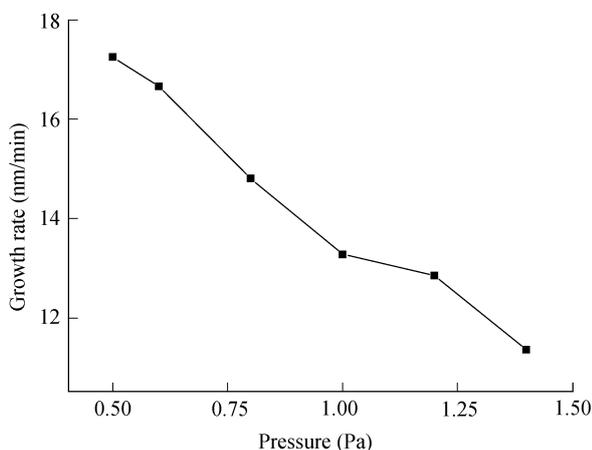
**Table 1** Deposition parameter

Parameters	Value
Ar flow rate (cm <sup>3</sup> /min)	16
O <sub>2</sub> flow rate (cm <sup>3</sup> /min)	4
Ti target sputtering power (W)	400
Base vacuum (Pa)	$1.0 \times 10^{-4}$
Working pressure (Pa)	0.5–1.4

## 2 Results and discussion

### 2.1 Growth rate analysis

Figure 1 shows the curve of the growth rate of TiO<sub>2</sub> films deposited at different working pressures. From Fig. 1, it can be seen that with increasing working pressure from 0.5 to 1.4 Pa, growth rate of films decreases gradually from 1.7 to 1.1 nm/min. This trend can be explained by the following reasons. First, working pressure of system controls mean free path of metastable Ar particles (Ar\*)



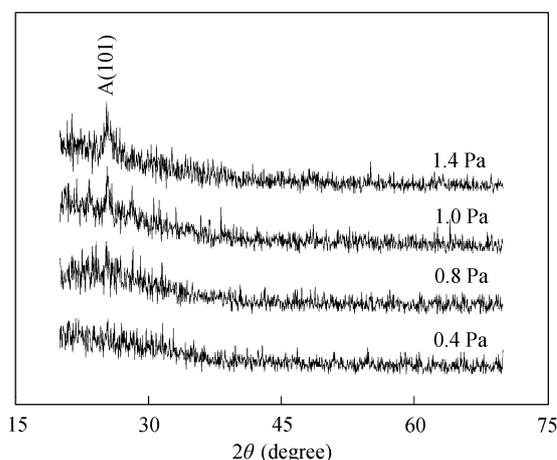
**Fig. 1** Growth rate at different working pressures.

and sputtering ions (Ar<sup>+</sup>) (Tang, 2003; Ding, 2007). As working pressure increases, the mean free paths of Ar\* and Ar<sup>+</sup> decrease and then the energy of Ar<sup>+</sup> decreases. Farther, the sputtering yield of target decreases. Second, mean free paths of Ti atoms and Ti ions decrease with increasing working pressure. Then the scattering cross section of Ti particles increases with increasing working pressure in the process of Ti particles incident to the growing surface of film.

### 2.2 XRD analysis

Figure 2 shows the XRD patterns of TiO<sub>2</sub> films deposited at different working pressures. From Fig. 2 it can be seen that the structure of TiO<sub>2</sub> films strongly depends on the working pressure. The XRD patterns of TiO<sub>2</sub> films deposited at low working pressures (0.4–0.8 Pa) display no obvious diffraction peak, which means that the films are amorphous structure. When working pressure is increased from 1.0 to 1.4 Pa, the XRD patterns of TiO<sub>2</sub> films exhibit only one diffraction peak at  $2\theta = 25.3^\circ$ , which means that the films are pure anatase (101) phase. No rutile phase is found in XRD patterns of all the films.

The rutile phase is a high temperature TiO<sub>2</sub> phase and therefore higher activation energy is needed for its formation comparing with the anatase phase (Sicha *et al.*, 2008). Figure 3 gives the Ti target self-bias at different working pressures. When other deposition parameters are seted constantly, the Ti target self-bias decreases monotonously with increasing working pressure. In sputtering theory, the Ti target self-bias variation influences the energy of particles sputtered from the target directly. Finally, the particles energy incidence to the film growth surface is affected by the Ti target self-bias variation (Tang, 2003; Ding, 2007). The particles energy is higher than need for anatase phase formation but lower than that for rutile phase from 0.5 to 0.8 Pa (Sicha *et al.*, 2008). Therefore, the film displays an amorphous structure. As mentioned above, higher working pressure leads to loss more particle energy by scattering between particles. Thus pure anatase phase TiO<sub>2</sub> films are deposited at high working pressure (1.4 Pa).



**Fig. 2** X-ray diffraction (XRD) patterns of TiO<sub>2</sub> films deposited at different working pressures.

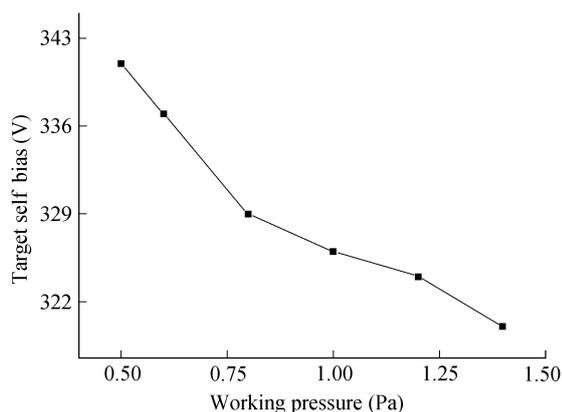


Fig. 3 Ti target self bias at different working pressures.

In order to see clearly the crystal structure of TiO<sub>2</sub> films, TiO<sub>2</sub> films deposited at different working pressures are thermal annealed at 500°C. Figure 4 shows the evolution of XRD patterns of annealed TiO<sub>2</sub> films. From Fig. 4, it can be seen that no rutile phase is found in all samples after thermal annealing treatment. The XRD patterns of TiO<sub>2</sub> films deposited at low working pressures (0.5–0.8 Pa) display diffraction peaks at  $2\theta = 25.3^\circ, 37.8^\circ, 48.1^\circ, 53.9^\circ,$  and  $62.7^\circ$ , which is generally attributed to anatase phase (101), (004), (200), (105), and (204) (00-021-1272 JCPDS card number), respectively. This phenomenon can be explained by the energy of incident Ti particles. Because the energy is lower than needed for rutile phase and the film contain little rutile phase nucleation, the XRD patterns of all films after thermal annealing treatment display no rutile phase. With a higher energy than needed for anatase phase, Ti particles incident to the film growth surface come into being anatase phase nucleation randomly. Thereby, after thermal annealing treatment, TiO<sub>2</sub> films deposited at low working pressure contain multi anatase crystallographic orientations. While with increasing working pressure, the incident Ti particles energy is decreased through scattering other particles, as discussed above. This results show that the films display the anatase (101) phase preferred

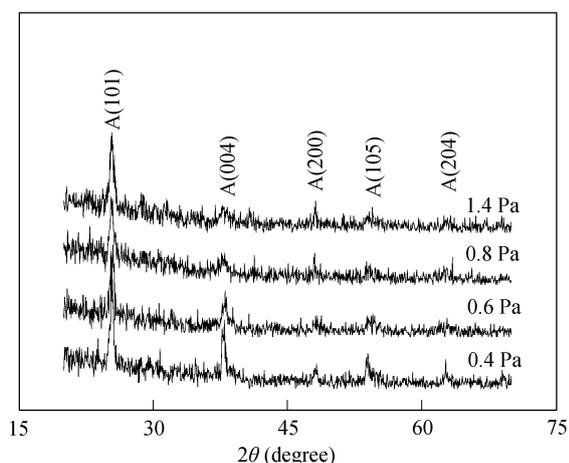


Fig. 4 XRD patterns of TiO<sub>2</sub> films after thermal annealing treatment at 500°C.

orientation. From Figs. 2 and 4, it is seen that both films before and after thermal annealing treatment, exhibit anatase (101) phase preferred orientation deposited at high working pressure (1.4 Pa).

### 2.3 Photocatalytic activity

The photocatalytic activity of TiO<sub>2</sub> film is characterized by the degradation rate of MO solution, which the degradation rate is the change of the solution concentration. The concentration of MO solution is determined by measuring the absorbency of solution at wavelength 485 nm. The absorbency of solution is determined by its absorption coefficient, thickness, and concentration. The absorption coefficient and thickness of solution are constant in all experiments. Therefore, the absorbency of solution represents the concentration of solution directly. Figure 5 shows the evolution of degradation rate for 200 nm as deposited TiO<sub>2</sub> films with increasing working pressure. From Fig. 5 it can be seen that the working pressure strongly influences the degradation rate of deposited TiO<sub>2</sub> films. The degradation rate of TiO<sub>2</sub> films improves from 17% to 33% when working pressure rises from 0.4 to 1.4 Pa. This trend can be interpreted as follows. First, anatase phase is referred to be the most photoactive phase. With the increase of working pressure, more anatase phase TiO<sub>2</sub> structure appeared, and then the degradation rate for MO solution is improved. Second, with working pressure increasing, the structure of TiO<sub>2</sub> film is transformed from amorphous into crystal. The morphological features of films change accordingly. Figure 6 displays the atomic force microscopy (AFM) surface morphologies of TiO<sub>2</sub> films with different working pressures. From Fig. 6 it can be seen clearly that the surface of TiO<sub>2</sub> films deposited at 0.4 Pa is much smoother than that at 1.4 Pa, which means the latter possess much more surface area contact with MO solution. Thus, the degradation rate of TiO<sub>2</sub> films improves when working pressure rises up. After UV irradiation for 5 h, MO solution could be degraded completely by TiO<sub>2</sub> film deposited at 1.4 Pa.

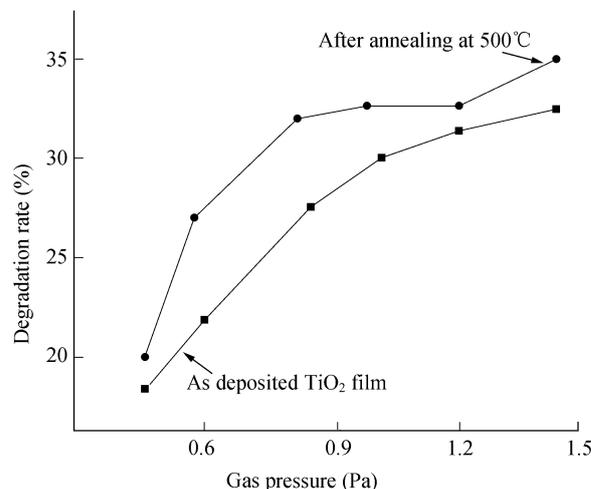
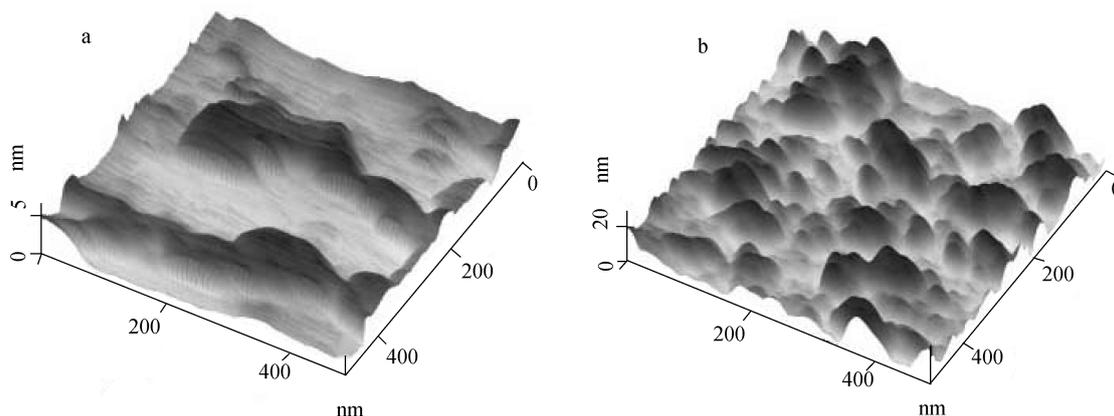


Fig. 5 Photocatalytic activity of TiO<sub>2</sub> film as deposited and after annealing at 500°C.



**Fig. 6** Atomic force microscopy (AFM) morphological images of TiO<sub>2</sub> films deposited at 0.4 Pa (a) and 1.4 Pa (b).

Comparing the degradation rate of TiO<sub>2</sub> film as deposited with that after thermal annealing treatment at 500°C (Fig. 5), it can be seen that thermal annealing treatment does not improve the photocatalytic activity of TiO<sub>2</sub> films and its evolution trend obviously. From photocatalytic activity analysis, the conclusion can be drawn that TiO<sub>2</sub> film with high photocatalytic activity can be deposited at room temperature by direct current pulse magnetron sputtering system, which corresponds well with the results above and possess particular feature in the system. Within this technology, TiO<sub>2</sub> films could be utilized in a wider application.

### 3 Conclusions

Experiments results show that TiO<sub>2</sub> films of high photocatalytic activity can be obtained by direct current pulse magnetron sputtering system at room temperature (substrate temperature < 100°C). Main conclusions can be summarized as follows. 1. The working pressure is key deposition parameter, which influences the target yield and kinetic energy of Ti particles. Photocatalytic activity of films attenuated with the incident Ti particles with high kinetic energy. 2. With the increase of working pressure, the structure of TiO<sub>2</sub> films transforms from amorphous phase to pure anatase(101) phase and the TiO<sub>2</sub> films deposited at high working pressure (1.4 Pa) display excellent photocatalytic activity. The degradation rate of MO solution is slightly different between deposited TiO<sub>2</sub> films and after annealing treatment TiO<sub>2</sub> films.

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